

Contents lists available at ScienceDirect

Journal of Hazardous Materials



journal homepage: www.elsevier.com/locate/jhazmat

Lead extraction from waste funnel cathode-ray tubes glasses by reaction with silicon carbide and titanium nitride

Pascal G. Yot^{a,*}, François O. Méar^{b,1}

 ^a Institut Charles Gerhardt Montpellier, UMR 5253 CNRS-UM2-ENSCM-UM1, CC 1503, Université Montpellier 2, Place Eugène Bataillon, 34095 Montpellier cedex 5, France
 ^b Unité de Catalyse et de Chimie du Solide, UMR 8181 CNRS-USTL-ENSCL, Université des Sciences et Techniques de Lille, 59652 Villeneuve d'Ascq cedex, France

ARTICLE INFO

Article history: Received 10 February 2009 Received in revised form 24 June 2009 Accepted 25 June 2009 Available online 3 July 2009

Keywords: CRT Lead silicate glass Chemical extraction Hazardous waste Heavy metal

1. Introduction

Since the end of the 90s, the re-use of waste electrical and electronic equipment (WEEE) also called e-waste, generated over the world is of great interest [1-7]. Considering that electrical and electronic equipment (EEE) is hazardous containing components and that the amount of WEEE has grown rapidly, the occidental governments have edited directives concerning their waste management and their re-use to ensure an environmental safety [8-10]. For example in the European Union (EU) WEEE represent about 7.5 million tons each year [11]. As a consequence, the computer monitors and TV sets containing cathode-ray tubes (CRTs) represent about 80% of the total electronic waste [12]. At the beginning of the decade it was estimated that 300,000 tons of e-waste ended up in landfill [13] and CRTs represent about one-third of electronics waste tonnage in the United States [14]. Considering that current CRT technology will become obsolete and replaced by thin-film transistor liquid display (TFT-LCD), a huge number of TV sets and computer monitors will be potentially disposed in landfill. CRTs are made of different glasses: the front part, usually known as the panel

ABSTRACT

As a possibility to clean waste CRT glass, treatment of lead-containing glass with a reducing agent, SiC or TiN, leads to a porous material containing metallic lead, Pb(0), located on the surface of the pore, and unreduced lead, Pb(II). The influences of reducing agent content, of the time, and at last of the temperature on lead reduction were analysed. Our investigations have pointed out significant differences as a function of the used reducing agent. CRT glass heat treated with SiC lead to less Pb(0), compared to TiN as shown by X-ray diffraction, and differential scanning calorimetry (DSC). It has been also evidenced that lead reduction occurs on randomized zones inside the sample leading to macroscopic lead beads inside glassy samples. XPS and XAS measurements were also carried out to investigate the local structure of lead and have evidenced a change of role of lead inside the glassy framework in function of the used conditions. © 2009 Elsevier B.V. All rights reserved.

which is a barium-strontium containing glass, the hidden parts inside the monitor, called funnel glass and the neck glass which are lead-rich silicate glass which envelopes the electron gun [15]. The amount of lead present in glasses used in cathode-ray tube could lead to an important social problem [12,15]. Some studies have demonstrated that the lead containing CRT glasses must be considered as hazardous waste, because of the well-known health effects of lead. Landfilling of CRT glasses, strictly restricted in the EU, could or should be solved by (i) recycling new CRTs glasses with the waste one (called closed loop system) [5,16]; (ii) manufacturing other outlets with CRT waste glasses (called open-loop system) [12,17–25]. However these solutions will be efficient if the leachability of heavy metal does not exceed the threshold limits; (iii) encapsulating lead [26]; or at least (iv) decontaminating the lead [27] or heavy metals [28,29] containing glasses. Development of techniques for the extraction of heavy metal ions from glass is also important in disposing waste glass used in electronic applications since these devices contain hazardous heavy metal elements, such as lead. These applications are the most revelling ones and more details on re-use of CRTs are given by Herat [7]. Previous studies were carried out in our laboratory [15,30–35] with investigations of the recycling of PbO-containing waste CRT glasses. Foam glass appeared to be a promising solution to re-use waste CRT glasses. The reduction of lead (+II) to metallic lead occurs to be a result of the reaction between SiC (or TiN) and the lead oxide contained in the glass, generating CO₂ (or N₂) within the framework [17,31-33] producing a porous cellular structure. It has been

^{*} Corresponding author. Tel.: +33 4 67 14 32 94; fax: +33 4 67 14 42 90.

E-mail addresses: pascal.yot@univ-montp2.fr (P.G. Yot), francois.mear@univ-lille1.fr (F.O. Méar).

¹ Tel.: +33 3 20 33 72 46; fax: +33 3 20 43 68 14.

^{0304-3894/\$ -} see front matter © 2009 Elsevier B.V. All rights reserved. doi:10.1016/j.jhazmat.2009.06.137

Table 1

Reducing agent contents, temperature and time used to determine the lead content in the samples for the various conditions (fixed parameters are underlined).

	Reducing agent content (wt.%)		Temperature of reaction (°C)	Time of reaction (min)
	SiC	TiN		
Effect of the reducing agent	$1 \rightarrow 17$		<u>850</u>	60
Effect of the temperature	5		700 ightarrow 1000	<u>60</u>
Effect of the time of reaction	<u>5</u>		<u>850</u>	10 ightarrow 180

demonstrated that metallic lead located on the surface of the pore, was obtained in different proportions depending on the elaboration parameters. In extreme conditions, i.e. high reducing agent content, high temperature and long time of reaction, it has been possible to extract lead from the initial glass. This work aimed to investigate the influence of reducing agent content (1-17 wt.%), time of reaction (10-180 min) and temperature of reaction $(700-1000 \,^{\circ}\text{C})$ on the lead extraction. The effects of elaboration parameters on the lead reduction process are also described. The local structure around unreduced lead has been studied by spectroscopic tech-



Fig. 1. XRD patterns obtained as a function of the reducing agent content for samples elaborated with SiC (a) and TiN (b). Reduced lead content evolution (c) obtained by DSC for various SiC and TiN contents (in wt.% of the initial lead (+II)).



Fig. 2. XRD patterns obtained as a function of the temperature for samples elaborated with SiC (a) and TiN (b). Reduced lead content evolution (c) obtained by DSC for various SiC and TiN contents (in wt.% of the initial lead (+II)).

niques in previous paper and will be used to discuss the results obtained here.

2. Materials and methods

2.1. Preparation of the samples

Waste funnel glass powder and reducing agent both in powder forms (SiC Aldrich 99.9% and TiN Alfa Aesar 99.8%), were mixed with a binder and then pressed at room temperature under 5 tons to get pellets with 40 mm of diameter and thickness of 6–8 mm. The average particle sizes of the different powders used were smaller than 63 μ m. All the pellets were treated in an electric furnace under air atmosphere in the heating conditions summarized in Table 1. In the whole conditions, the samples were removed from the furnace and allowed to reach room temperature in 'free fall'.

2.2. Lead reduction reactions

Based on the previous works [31], the reaction processes of interest for the two reducing agents can be described with the following equations:

$$2PbO(s) + 2TiN(s) + O_2(g) \leftrightarrow 2Pb(s) + 2TiO_2(s) + N_2(g)$$

 $2PbO(s) + SiC(s) + O_2(g) \leftrightarrow 2Pb(s) + SiO_2(s) + CO_2(g)$

where TiO_2 and SiO_2 are in a vitreous phase [32].

2.3. Lead content determination

X-ray diffraction patterns were determined for samples with various quantities of the reagent grade SiC or TiN using a PANalytical X'PERT diffractometer. Each sample was grinded to a fine powder (50 µm) and was introduced into a sealed silica capillary tube of 0.3 mm diameter. The X-ray diffraction (XRD) patterns of the whole samples were collected for 2θ from 10° to 60° using CuK_{α} radiation $(\lambda = 1.5604 \text{ Å}).$

The variation of the lead content in the samples was measured using the areas of the differential scanning calorimetry (DSC) peaks corresponding to the lead melting point (327 °C). The measurements were carried out on a dynamic heat-flux differential scanning calorimeter (NETZSCH DSC 200) in a sealed aluminium crucible using a heating rate of $5 \circ C/min$ from 20 to 400 $\circ C$.

3. Results and discussion

3.1. Effect of the reducing agent content

The influence of the reducing agent on the lead content was studied at constant temperature (850 °C) with a reaction time of 60 min. X-ray diffraction patterns obtained for 1–17 wt.% of reducing agent are shown in Fig. 1(a) and (b). Sharp diffraction peaks due to the crystallization of the lead are seen, in addition to a broad diffraction peak revealing the conservation of an amorphous character. All the crystalline peaks are identified to belong to Pb (PCPDF 04-0686) and residual reducing agent SiC (PCPDF 49-1428, Fig. 1(a)) or TiN (PCPDF 08-0418, Fig. 1(b)) phases. In addition, it is seen that the fraction of Pb(0) phase increases significantly (revealed by the height of the diffraction peaks) with increase the initial content of SiC or TiN.

The quantitative estimation of the reduced lead content was carried out by DSC and presented in Fig. 1(c). The variation of the lead (0) content as a function of reducing agent content in case of samples reduced with SiC was found to be linear up to 9 wt.%, whereas the variation with TiN was found to be exponential (Fig. 1(c)). Thus with 9 wt.% TiN using, it would appear that 20 wt.% of the lead initially present in the glass is reduced to the lead metal form after heat treatment, for only 12 wt.% with SiC using.

3.2. Effect of the temperature

The effect of temperature was investigated using a fixed SiC or TiN reducing agent content of 5 wt.% and a synthesis time of 60 min. The corresponding X-ray diffraction patterns, shown in Fig. 2(a) and (b), indicate, as mentioned before, the presence of both non-crystalline phase, metallic lead, and residual reducing agent. Fig. 2(c) shows the results of the DSC measurements of the lead quantity as a function of the temperature. For both SiC and TiN, the trends are guite similar. The two curves present a maximum of reduced lead at 950 and 925 °C for SiC and TiN, respectively. At higher temperatures, the reduced lead quantity found by DSC decreases with the temperature. This significant decrease is the result of the coalescence of lead granules, as observed by microscopy, inside the material (Fig. 3). On the effect of the high temperature, macroscopic granules of lead have coalesced inside the pellet (see the arrows on the picture in Fig. 3) to give an inhomogeneous material with lead rich zone and poor lead zone. The powder used for the XRD and DSC analysis, came from the zone lacking of lead (0) in the sample. The higher is the temperature of reaction, the largest are lead granules and then the lowest is the lead (II) content inside the glassy frameworks. Hence, the probability to use for the DSC measurements poor lead containing zone increases, explaining the shape of the two curves in Fig. 2(c). We can

lead granules 10 mm Fig. 3. Photograph of the lead granules obtained by coalescence inside the material at high temperature.

therefore assume that more than 20 and 40 wt.%, could be obtained with increasing the temperature, for TiN and SiC, respectively. Based on these results, increasing both the temperature and the reducing agent content, could lead to 'phase separated samples': a pure lead metallic phase and a more or less poor lead silicate glass phase.

3.3. Effect of the time of reaction

The effect of the reaction time on the lead content was investigated at 850 °C with 5 wt.% content of SiC and TiN reducing agent. Once again, the corresponding X-ray diffraction patterns indicate, the presence of non-crystalline phase, metallic lead and remaining reducing agent. The lead content increases with increasing the reaction time and reaches to a maximum at 20 wt.% for both, SiC and TiN using (Fig. 4(c)). Such behaviour is similar whatever the nature of the reducing agent.

The metallic lead amounts obtained by the investigated process are lower than those obtained through pyrovacuum process by Chen et al. [27] using carbon. As is well known, carbon is one of the best reducing agent used in the metallurgy for steal synthesis. Considering this point, it gives a response on the lower lead quantities obtained in our case. Analysing the aforementioned results, the reduced lead content could be more important increasing temperature and/or reducing agent content. Nevertheless, as the reaction occurs in air atmosphere, oxygen pressure will be a limitative parameter for the lead reduction.

3.4. Evolution of the local structure around lead

The local environment of the unreduced lead was then investigated using X-ray photoelectron spectroscopy (XPS) [33,36] and X-ray absorption spectroscopy (XAS) [37]. In our previous work [33] XPS spectra obtained for sample with 5 wt.% of TiN treated at 850 $^\circ$ C compared to PbSiO₃, respectively. XPS spectra of four Pb 4f XPS peaks were observed: two symmetric XPS peaks due to Pb 4f_{7/2} and Pb $4f_{5/2}$. The XPS results confirm the presence of two types of lead in the foam glass:

- Pb^{2+} ions with binding energies of Pb $4f_{7/2}$ (143.6 eV) and Pb $4f_{5/2}$ (138.6 eV), respectively corresponding to those of PbSiO₃ and
- Pb^0 metallic lead with binding energies Pb $4f_{7/2}$ (141.5 eV) and Pb 4f_{5/2} (136.6 eV).

As is well known, PbSiO₃ consists of infinite metasilicate chains interconnected by Pb²⁺ ions. The lead oxide is a glass modifier in CRT glasses, so its introduction results in the depolymerization of the



silica network [36]. Nevertheless, the relative position of the main transition (Pb 4f) for lead (II) is not significantly influenced by the variation of the reducing agent content [33]. The three dimensional framework of SiO₄ tetrahedra is maintained as lead oxide is present at low concentrations (20–22 wt.% corresponding to 2–3 mol.%) that vary slightly from one sample to another. The results show that the spectra obtained for a higher temperature of reaction are the same as those obtained for different reducing agent content. Lead (0) and (II) binding energies corresponding to Pb 4f_{7/2} and Pb 4f_{5/2} have been found at similar values showing that lead (II) remains present in the same environment. Nevertheless, the relative area of the peaks corresponding to lead metal increases, while the relative area of the Pb 4f peaks corresponding to lead in a PbSiO₃ environment decreases (Fig. 5).

In some previous works, the influence of the hydrogen reduction on the structure of lead silicate glasses was studied by EXAFS [38,39]. It has been demonstrated that the coordination number of unreduced lead decreases without change in Pb–O bond length which remains equal to 2.23 Å. In our case EXAFS has been also used and it allowed us to measure the influence of the reducing agent content. Lead–oxygen coordination number was determined using EXAFS amplitude which is proportional to a product of the coordination number (*N*) and the amplitude reduction factor S_0^2 which has been fixed to 0.75 [40]. Fig. 6 summarizes the evolution of the coordination number as a function of the reducing agent content. In the both cases lead coordination number decreases increasing the reducing agent content. Nevertheless the decrease is more pronounced in case of TiN than SiC. Lead–oxygen coordination number



Fig. 4. XRD patterns obtained as a function of the time of reaction for samples elaborated with SiC (a) and TiN (b). Reduced lead content evolution (c) obtained by DSC for various SiC and TiN contents (in wt.% of the initial lead (+II)).



Fig. 5. Pb 4f core level spectra of the foam glasses (a) and PbSiO₃ glass (b).



Fig. 6. Evolution of the average coordination of Pb–O as a function of the reducing agent content.

dramatically falls from 3 to 1 for TiN instead of 3 to 2 in case of SiC [37]. In both cases the Pb–O bond length remains around (2.23 Å); which is the same distance found in initial funnel glass [41]. These results are then consistent with results previously reported [38,39] and show the change of role of lead inside the glass under the influence of the reduction.

4. Conclusions

The influence of the lead metal content on the parameters as a function of reducing agent content, time and temperature on the foam glass synthesis has been studied using lead-containing waste CRT glasses (funnel) and two reducing agents, SiC and TiN. The treatment of glasses in adapted conditions leads to more and more friable and porous materials containing more Pb(0) granules and less unreduced Pb(II).

In this study, it has been shown that lead metal content is function of the experimental parameters. In fact, the higher the experimental conditions (reducing agent content, temperature and time) the higher the lead metal content. As it has been demonstrated by XRD, DSC and XAS, TiN treatment of CRT glasses gives less reduction of Pb(II) to Pb(0) than SiC treatment. As a result it could be possible to extract nearly whole the lead contained in the funnel glass (initial content 20–22 wt.% of lead oxide) in extreme condition of temperature, time of reaction and/or reducing agent content.

Function of the reducing agent content, 12 and 20 wt.% of total lead reduced (in foam glass synthesized 60 min at 850 °C) to metal form can be extracted with 9 wt.% SiC and 9 wt.% TiN used, respectively. Function of the temperature, 40 and 20 wt.% of total lead reduced (in foam glass synthesized 60 min at 950 °C) to metal form can be extracted with 5 wt.% SiC and 5 wt.% TiN used, respectively. And finally, function of time of reaction, 20 wt.% of total lead reduced (in foam glass synthesized 180 min at 850 °C) to metal form can be extracted with 5 wt.% SiC and 5 wt.% TiN used, respectively.

The chemical process has been studied by various spectroscopic methods. No variation of Pb–O bond length was found with XAS. In contrast the reduction process induces a dramatic decrease of number of coordination of lead–oxygen leading to a change in role of the unreduced lead in the glassy frameworks. This process allows both to obtain foam glass (a promising insulator material with thermal, electric or phonic insulator properties) and also to obtain a glass with low lead content. Nevertheless if SiC and TiN reducing agents could be used to extract lead from funnel CRT glass this process could be limited by its cost.

Acknowledgements

Authors would like to thank ANRT (France), IBM (France) and APF (France) for their financial support and kind assistance.

References

- Recytube, Integrated recycling of end of life cathode-ray tube glass, twoyear project partially under the European Commission's, BRITE-EURAM III, BRPR960246, 1996–1999.
- [2] N. Menad, Cathode-ray tube recycling, resources, Conserv. Recycling 26 (1999) 143–154.
- [3] C.H. Lee, S.L. Chang, K.M. Wang, L.C. Wen, Management of scrap computer recycling in Taiwan, J. Hazard. Mater. 73 (2000) 209–220.
- [4] E. Doring, TV glass recycling in Europe-description of the situation and possibilities, 1–2 (2002).
- [5] ICER (Industrial Council for Electronic Equipment, Recycling), Material recovery from waste cathode-ray tubes (CRTs), in: Creating Markets for Recycled Resources, Waste & Resource Action Programme (WRAP), Oxon, UK, 2004.
- [6] A.S. Smith, Recycled CRT panel glass as an energy reducing fluxing body additive in heavy clay construction products, in: Creating Markets for Recycled Resources, Waste & Resource Action Programme (WRAP), Oxon, UK, 2004.
- [7] S. Herat, Recycling of cathode ray tubes (CRTs) in electronic waste, Clean 36 (2008) 19–24.
- [8] USEPA (United State Environmental Protection Agency), Analysis of five community consumer/residential collections: end of life electronic and electrical equipment, Report, Washington DC, USA, 1999.
- [9] EU Directive, Directive 2003/108/EC of the European parliament and of the council of 8 December 2003 amending Directive 2002/96/EC on waste electrical and electronic equipment (WEEE), in: Official Journal of the European Communities, L 345, 31/12/2003, 2003, pp. 106–107.
- [10] United State Government Accountability Office, Electronic Waste: Strengthening the Role of the Federal Government in Encouraging Recycling and Reuse, Report GA0-06-47, USA, 2005.
- [11] M. Marshall, J. Henderson, New approaches to the challenge of CRT recycling, in: T. Telford (Ed.), Recycling and Re-use of Glass Cullet, 2001, pp. 75–83.
- [12] F. Andreola, L. Barbieri, A. Corradi, I. Lancellotti, R. Falcone, S. Hreglich, Glassceramics obtained by the recycling of end of life cathode-ray tubes glasses, Waste Manage. 25 (2005) 183–189.
- [13] C. Gable, B. Shireman, Computer and electronic product stewardship: are we ready for the challenge? in: Environmental Quality Management, Axioun Books, 2001, pp. 35–45.
- [14] T.G. Townsend, S. Musson, Y.C. Jang, I.H. Chung, Characterization of lead leachability from cathode-ray tubes using the toxicity characteristic leaching procedure, Report #99-5, Florida Center for Solid and Hazardous Waste Management, Gainesville, Florida, USA, 1999.

- [15] F. Méar, P. Yot, M. Cambon, M. Ribes, The characterization of waste cathode-ray tube glasses Waste Manage. 26 (2006) 1468–1476.
- [16] J. Piers, J. Peelen, US Patent 5,725,627 (1998).
- [17] C. Garnier, P. Verdier, J. Razafindrakoto, Y. Laurent, WO Patent 9831639 (1998).
- [18] J. M. Turmel, J. Rocherulle, P. Grange, J. Razafindrakoto, P. Verdier, Y. Laurent, WO Patent 0871520 (1998).
- [19] E. Bernardo, G. Scarini, S. Hreglich, Mechanical properties of metal-particulate lead-silicate glass matrix composites obtained by means of powder technology, J. Eur. Ceram. Soc. 23 (2003) 1819–1827.
- [20] F. Andreola, L. Barbieri, A. Corradi, I. Lancellotti, CRT glass state of art. A case study: recycling in ceramic glazes, J. Eur. Ceram. Soc. 27 (2007) 1623–1629.
- [21] E. Bernardo, R. Castallan, S. Hreglich, Al₂O₃-platelet reinforced glass matrix composites from a mixture of wastes, J. Mater. Sci. 42 (2007) 2706–2711.
- [22] E. Bernardo, R. Castallan, S. Hreglich, Sintered glass ceramics from mixtures of wastes, Ceram. Int. 33 (2007) 27–33.
- [23] E. Bernardo, G. Scarinci, S. Hreglich, G. Zangiacomi, Effect of time and furnace atmosphere on the sintering of glasses from dismantled cathode ray tubes, J. Eur. Ceram. Soc. 27 (2007) 1637–1643.
- [24] F. Andreola, L. Bardieri, E. Karamanova, I. Lancellotti, M. Pelino, Recycling of CRT panel glass as fluxing agent in the porcelain stoneware tile production, Ceram. Int. 34 (2008) 1289–1295.
- [25] Z. Matamoros-Veloza, J.C. Rendón-Angeles, K. Yanagisawa, M.A. Cisneros-Guerrero, M.M. Cisneros-Guerrero, L. Aguirre, Preparation of foamed glasses from CRT TV glass by means of hydrothermal hot-pressing technique, J. Eur. Ceram. Soc. 28 (2008) 739–745.
- [26] D. Kim, M. Quinlan, T.F. Yen, Encapsulation of lead from hazardous CRT glass waste using biopolymer cross-linked concrete systems, Waste Manage. 29 (2009) 321–328.
- [27] M. Chen, F.-S. Zhang, J. Zhu, Lead recovery and the feasability of foam glass production from funnel glass of dismantled cathode ray tube through pyrovacuum process, J. Hazard. Mater. 161 (2009) 1109–1113.
- [28] D. Chen, M. Masui, H. Miyoshi, T. Akai, T. Yazawa, Extraction of heavy metals ions from waste colored glass through phase separation, Waste Manage. 26 (2006) 1017–1023.
- [29] K.-D. Kim, Y.-H. Kim, Voltametric approach to redox behaviour of various elements in cathode ray tube glass melts, J. Non-Cryst. Solids 354 (2008) 553–557.

- [30] P. Yot, M. Assadinik, M. Cambon, B. Meyrand, B. Liautard, M. Ribes, Recycling of end of life cathode-ray tubes-synthesis and characterization of foam glass, in: Proceedings of National workshop of research in Technological Institute of University, tome 1, Publications of University of Saint-Etienne, France, 2001, pp. 312–324.
- [31] F. Méar, Study of foam glass elaborated from waste cathode-ray tubes (CRT) lead oxide containing glasses: elaboration, physical and chemical characterizations and applications, Ph.D. Thesis, University Montpellier 2, Montpellier, France, 2004.
- [32] F. Méar, P. Yot, M. Cambon, M. Ribes, Properties and structural characterization of foam glass elaborated from cathode-ray tube, Adv. Appl. Ceram. 104 (2005) 123–130.
- [33] F. Méar, P. Yot, M. Cambon, M. Ribes, The changes in lead silicate glasses induced by the addition of a reducing agent (TiN or SiC), J. Non-Cryst. Solids 351 (2005) 3314–3319.
- [34] F. Méar, P. Yot, M. Ribes, Effects of temperature, reaction time and reducing agent content on the synthesis of macroporous foam glasses from waste funnel glasses, Mater. Lett. 60 (2006) 929–934.
- [35] F. Méar, P. Yot, R. Viennois, M. Ribes, Mechanical behaviour and thermal and electrical properties of foam glass, Ceram. Int. 33 (2007) 543– 550.
- [36] P.W. Wang, L. Zhang, Structural role of lead in lead silicate glasses derived from XPS spectra, J. Non-Cryst. Solids 194 (1996) 129–134.
- [37] P.G. Yot, F.O. Méar, Influence of AIN, TiN and SiC reduction on the structural environment of lead in waste cathode-ray tubes glass: an X-ray absorption spectroscopy study, J. Phys. Condens. Matter 21 (2009) 285104.
- [38] A. Witkowska, J. Rybicki, K. Trzebiatowski, A. Di Cicco, M. Minicucci, Influence of hydrogen reduction on the structure of PbSiO₃, J. Non-Cryst. Solids 276 (2000) 19–26.
- [39] A. Witkowska, J. Rybicki, A. Di Cicco, Structure of partially reduced xPb0(1-x)SiO₂ glasses: combined EXAFS and MD study, J. Non-Cryst. Solids 35 (2005) 380–393.
- [40] E.A. Stern, in: D.C. Konningsberger, R. Prins (Eds.), X-Ray Absorption, John Wiley & Sons, New York, 1988, pp. 1–52.
- [41] F.O. Méar, P.G. Yot, A.V. Kolobov, M. Ribes, M.-F. Guimon, D. Gonbeau, J. Non-Cryst. Solids 353 (2007) 4640–4646.